

# Influence of Zn and Ni substitutions for Mg on dielectric properties of $(\text{Mg}_{4-x}\text{M}_x)(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$ ( $M = \text{Zn}$ and $\text{Ni}$ ) solid solutions

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## Abstract

The effects of  $M$  substitution for Mg and Sb substitution for Nb in the  $(\text{Mg}_{4-x}\text{M}_x)(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $M = \text{Zn}$  and  $\text{Ni}$ ) ( $x = 0-2$ ,  $y = 0-1.5$ ) solid solutions on the microwave dielectric properties were investigated in this study. The limits of both  $(\text{Mg}_{4-x}\text{M}_x)\text{Nb}_2\text{O}_9$  ( $M = \text{Zn}$  and  $\text{Ni}$ ) solid solutions were approximately  $x = 2$ , whereas that of  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions was approximately  $y = 1.5$ . As for the Ni and Zn substitutions for Mg, the quality factor,  $Q \cdot f$  values of the  $(\text{Mg}_{4-x}\text{M}_x)\text{Nb}_2\text{O}_9$  ( $M = \text{Zn}$  and  $\text{Ni}$ ) solid solutions decreased from 192 268 to 28 400 GHz with increasing composition  $x$  from 0 to 2. On the other hand, the  $Q \cdot f$  values of the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions were remarkably improved by the Sb substitution for Nb; the highest  $Q \cdot f$  value of 285 423 GHz was obtained at  $y = 1$ . It was found that an increase in the  $Q \cdot f$  values of  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions may related to the grain size decreased which is opposite to general results and this relation is consistent with that of  $\text{Al}_2\text{O}_3$ .

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**Keywords:** Dielectric properties; Grain size; Niobates; Powders–solid state reaction

## 1. Introduction

With the recent progress in microwave components, the development of high- $Q$  materials with a variety of dielectric constants ( $\epsilon_r$ ) and small temperature coefficient of resonant frequency ( $\tau_f$ ) have been required for application as a dielectric resonator, a high-temperature superconductor (HTSC) filter and as the substrate for integrated circuits. The  $\text{Al}_2\text{O}_3$  substrate is widely used for the integrated circuits because of its suitable dielectric properties for the applications ( $\epsilon_r \approx 10$ ,  $\tan \delta < 10^{-4}$ ).<sup>1</sup> Thus, the development of new dielectric ceramics with low dielectric loss, which are comparable to that of the  $\text{Al}_2\text{O}_3$  substrate, is required. The  $\text{Mg}_4\text{Nb}_2\text{O}_9$  compound with  $\tan \delta$  value lower than  $10^{-4}$  which has a corundum-type crystal structure,<sup>2</sup> have been developed in our previous work.<sup>3</sup> In this study, the effects of  $M$  ( $M = \text{Zn}$  and  $\text{Ni}$ ) substitutions for Mg and the Sb substitution for Nb on the microwave dielectric properties were investigated.

Thus, the  $(\text{Mg}_{4-x}\text{M}_x)(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $M = \text{Zn}$  and  $\text{Ni}$ ) solid solutions were synthesized and the relationships among crystal structure, microstructure and microwave dielectric properties of the solid solutions were discussed.

## 2. Experimental

The  $(\text{Mg}_{4-x}\text{M}_x)(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $M = \text{Zn}$  and  $\text{Ni}$ ) solid solutions were synthesized by the conventional solid-state reaction method. High purity ( $> 99.9\%$ )  $\text{MgO}$ ,  $\text{ZnO}$ ,  $\text{NiO}$ ,  $\text{Nb}_2\text{O}_5$  and  $\text{Sb}_2\text{O}_5$  powders weighed on the basis of the stoichiometric composition were mixed and calcined at  $1000^\circ\text{C}$  for 20 h in air. These calcined powders were milled and mixed with an organic binder, and then pressed into a pellet with 12 mm in diameter and 7 mm in thickness under the pressure of 100 MPa. Subsequently, these pellets were sintered at the various temperatures ranging from  $1200$  to  $1500^\circ\text{C}$  for 10 h in air. The phases of the samples synthesized were identified by X-ray powder diffraction (XRPD). The microstructure was investigated by using the scanning electron microscopy

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(SEM) and the energy dispersive X-ray (EDX). The lattice parameters and crystal structure of the samples were evaluated according to the Rietveld analysis<sup>4,5</sup> and the least squares method. The microwave dielectric properties were determined by Hakki and Coleman's method.<sup>6</sup>

### 3. Results and discussion

#### 3.1. $(Mg_{4-x}M_x)Nb_2O_9$ ( $M = Zn$ and $Ni$ ) solid solutions

Figs. 1 and 2 show the XRPD patterns of  $(Mg_{4-x}Zn_x)Nb_2O_9$  and  $(Mg_{4-x}Ni_x)Nb_2O_9$  solid solutions, respectively. The XRPD results show that the  $(Mg_{4-x}Zn_x)Nb_2O_9$  ceramics were single phase at the compositions  $x$  ranging from 0 to 2, whereas three phases, i.e.,  $ZnO$ ,  $ZnNb_2O_6$  and  $Zn_3Nb_2O_8$ , coexisted at  $x=4$  as shown in Fig. 1(c). The XRPD results of  $(Mg_{4-x}Ni_x)Nb_2O_9$  ceramics showed the presence of  $NiO$  and  $NiNb_2O_6$  at  $x=4$  instead of  $Ni_4Nb_2O_9$  compound. These results suggest that both the limits of  $(Mg_{4-x}Zn_x)Nb_2O_9$  and the  $(Mg_{4-x}Ni_x)Nb_2O_9$  solid solutions may be the compositions between  $x=2$  and  $x=4$ . The influence of the differences in ionic radii of  $Mg^{2+}$ ,  $Zn^{2+}$  and  $Ni^{2+}$  ions on the crystal structure, the lattice parameters of the  $(Mg_{4-x}M_x)Nb_2O_9$  solid solutions are determined as shown in Fig. 3. Further details on the crystal structure of the ceramics at  $x=0$ , i.e., that of the  $Mg_4Nb_2O_9$  compound were reported by Kumada et al.<sup>2</sup> It has a trigonal structure (S.G.  $P\bar{3}c1$ ) with the lattice parameters,  $a=5.1612\text{\AA}$  and  $c=14.028\text{\AA}$ . The lattice parameters of the ceramics at  $x=0$  obtained in this study coincided with those values. The lattice

parameters,  $a$  and  $c$ , of the  $(Mg_{4-x}Zn_x)Nb_2O_9$  solid solutions were increased linearly with increasing the compositions  $x$  ranging from 0 to 2 and then these values substantially saturated at the composition  $x=2$ . Moreover, in the case of  $Ni$  substitution for  $Mg$ , the lattice parameter  $a$  was decreased with increasing the compositions  $x$  while that of  $c$  was increased as shown in Fig. 3. The variations in the lattice parameters of each sample depend on the differences in the ionic radii of  $Mg^{2+}$  (0.72\AA),  $Zn^{2+}$  (0.74\AA) and  $Ni^{2+}$  (0.68\AA) ions under the same coordination number (C.N.=6 for the  $MgO_6$  and  $MO_6$  octahedra). The microwave dielectric properties of  $(Mg_{4-x}M_x)Nb_2O_9$  ( $M = Zn$  and  $Ni$ ) ( $x=0-2$ ) solid solutions are listed in Table 1. The  $Q \cdot f$  values of the  $(Mg_{4-x}M_x)Nb_2O_9$  ( $M = Zn$  and  $Ni$ ) ceramics deteriorated with the  $M$  substitutions for  $Mg$ . The  $\tau_f$  values of the  $(Mg_{4-x}Zn_x)Nb_2O_9$  solid solutions were decreased from  $-70.5$  to  $-95.8$  ppm/ $^{\circ}C$ , and those of  $(Mg_{4-x}Ni_x)Nb_2O_9$  solid solutions were slightly increased from  $-70.5$  to  $-66.5$  ppm/ $^{\circ}C$ .

#### 3.2. $Mg_4(Nb_{2-y}Sb_y)O_9$ ( $y=0-1.5$ ) solid solutions

From the XRPD results of the  $Sb$ -substituted  $Mg_4(Nb_{2-y}Sb_y)O_9$  ( $y=0-1.5$ ) solid solutions, no secondary phase was detected in the compositions ranging from 0 to 1.5. The lattice parameters and unit cell volumes of the  $Mg_4(Nb_{2-y}Sb_y)O_9$  solid solutions were decreased linearly because the ionic radius of  $Sb^{5+}$  is smaller than that of  $Nb^{5+}$  and the lattice parameters in the compositions  $y$  ranging from 0 to 1.5 were shown in Table 2. The microwave dielectric properties of the  $Mg_4(Nb_{2-y}Sb_y)O_9$  solid solutions are shown in Table 3. The  $\epsilon_r$  values of the  $Mg_4(Nb_{2-y}Sb_y)O_9$  solid solutions varied from 12.4 to 10.0. The  $Q \cdot f$  values of the  $Mg_4(Nb_{2-y}Sb_y)O_9$  solid solutions were extremely increased from 192 268 to 285 423 GHz, depending on the compositions  $y$ . The highest value, i.e.,  $Q \cdot f=285$  423 GHz, was obtained at  $y=1$ . Thus, it was found that the  $Sb$  substitution for  $Nb$  is effective in increasing the  $Q \cdot f$  values of the  $Mg_4(Nb_{2-y}Sb_y)O_9$  solid solutions.

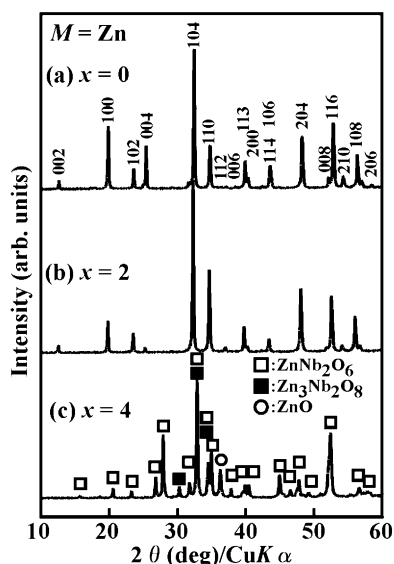


Fig. 1. XRPD patterns of the  $(Mg_{4-x}Zn_x)Nb_2O_9$  solid solutions.

Table 1  
Dielectric properties of  $(Mg_{4-x}M_x)Nb_2O_9$  solid solutions

| $M$ | $x$ | $D_r$ % | $\epsilon_r$ | $Q \cdot f$ (GHz) | $\tau_f$ (ppm/ $^{\circ}C$ ) |
|-----|-----|---------|--------------|-------------------|------------------------------|
| Zn  | 0   | 92.8    | 12.4         | 192 268           | -70.5                        |
|     | 0.5 | 95.0    | 13.7         | 105 383           | -76.9                        |
|     | 1.0 | 93.8    | 14.2         | 76 273            | -84.0                        |
|     | 1.5 | 94.8    | 15.3         | 61 612            | -86.7                        |
|     | 2.0 | 89.8    | 15.4         | 52 241            | -95.9                        |
| Ni  | 0.5 | 93.7    | 12.8         | 84 270            | -68.3                        |
|     | 1.0 | 92.6    | 12.6         | 44 025            | -69.1                        |
|     | 1.5 | 90.3    | 12.1         | 33 447            | -66.5                        |
|     | 2.0 | 92.3    | 11.1         | 28 440            | -69.6                        |

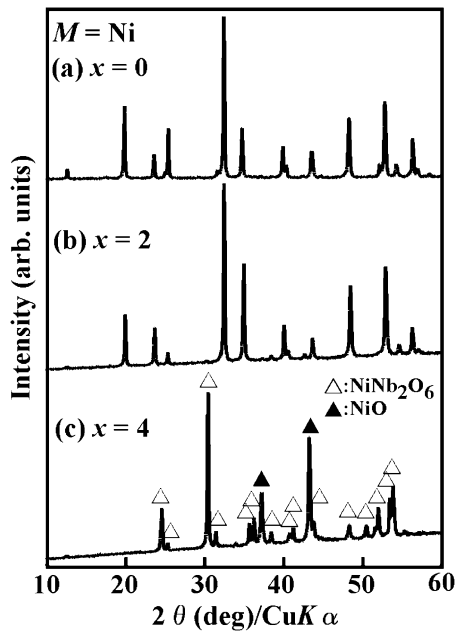


Fig. 2. XRPD patterns of the  $(\text{Mg}_{4-x}\text{Ni}_x)\text{Nb}_2\text{O}_9$  solid solutions.

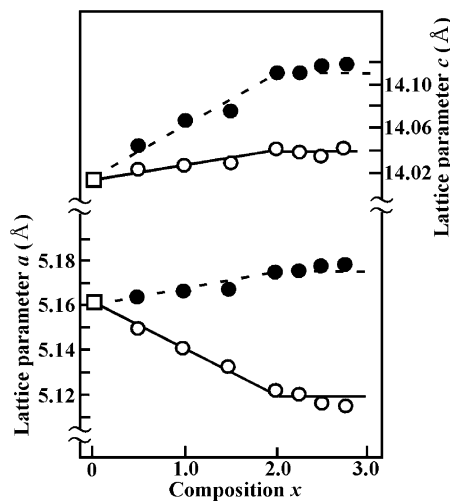


Fig. 3. The lattice parameters of the  $(\text{Mg}_{4-x}\text{M}_x)\text{Nb}_2\text{O}_9$  solid solutions.

However, The samples with compositions higher than  $y=1$ , the microwave dielectric properties of the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions were decreased because of the Sb vaporization. Thus, in order to clarify the relationship between the microstructure and an increasing in the  $Q \cdot f$  values caused by the Sb substitution for Nb. The FE-SEM micrographs (SEI) of  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $y=0-1$ ) solid solutions sintered at  $1400^\circ\text{C}$  for 10 h in air are shown in Fig. 4. Though the grain sizes of the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions were decreased with increasing compositions  $y$ , the formation of porosity and impurities were not observed in the

Table 2  
Lattice parameters and unit cell volumes of  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $y=0-1.5$ ) solid solutions

| $y$  | Lattice parameter ( $\text{\AA}$ ) |            | Unit cell volume ( $\text{\AA}^3$ ) |
|------|------------------------------------|------------|-------------------------------------|
|      | $a$                                | $c$        |                                     |
| 0    | 5.1636(3)                          | 14.0273(6) | 323.90                              |
| 0.25 | 5.1644(2)                          | 14.0200(4) | 323.71                              |
| 0.5  | 5.1638(2)                          | 14.0114(4) | 323.56                              |
| 0.75 | 5.1635(1)                          | 14.0018(3) | 323.34                              |
| 1.0  | 5.1642(1)                          | 13.9952(3) | 323.30                              |
| 1.5  | 5.1653(1)                          | 13.9860(2) | 322.83                              |

Table 3  
Dielectric properties of  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $y=0-1.5$ ) solid solutions

| $y$  | $D_r$ (%) | $\epsilon_r$ | $Q \cdot f$ (GHz) | $\tau_f$ (ppm/ $^\circ\text{C}$ ) |
|------|-----------|--------------|-------------------|-----------------------------------|
| 0.25 | 96.2      | 12.2         | 164854            | -66.5                             |
| 0.5  | 97.2      | 11.9         | 186496            | -69.2                             |
| 0.75 | 96.9      | 11.07        | 227904            | -69.9                             |
| 1.0  | 93.0      | 10.0         | 285423            | -66.2                             |
| 1.5  | 85.0      | 4.5          | 54686             | -49.9                             |

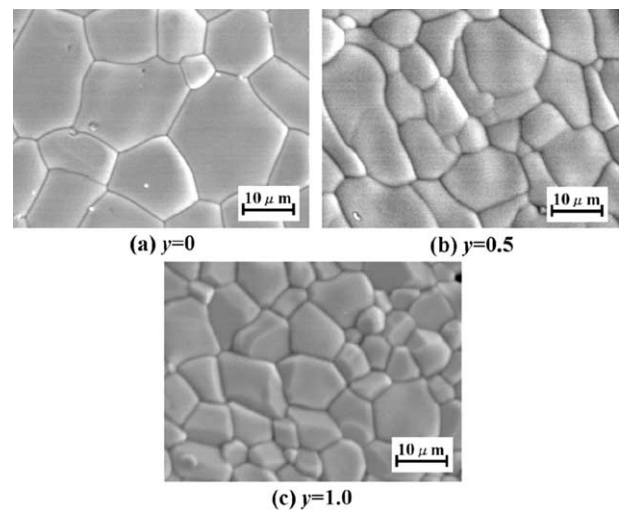


Fig. 4. FE-SEM micrographs (SEI) of the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $y=0-1$ ) solid solutions at (a)  $y=0$ , (b)  $y=0.5$  and (c)  $y=1.0$  sintered at  $1400^\circ\text{C}$  for 10 h.

compositions  $y$  ranging from 0 to 1. In general, it is known that the  $Q \cdot f$  values of the ceramics strongly depend on the ordering, morphological changes such as porosity, grain size, impurities and grain boundary.<sup>7</sup> As for the grain size of the other materials it has been observed that as grain size increases the loss is reduced, in contrast to the results of  $\text{Al}_2\text{O}_3$ <sup>7</sup> and the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions presented here. Fig. 5 is the plot of the  $\tan\delta$  and average grain size against its compositions  $y$ . Both variations in the grain size and the  $\tan\delta$  exhibit the similar tendencies with increasing the compositions  $y$ . Thus, it was found that the decrease in the grain size relates with the increasing  $Q \cdot f$  values.

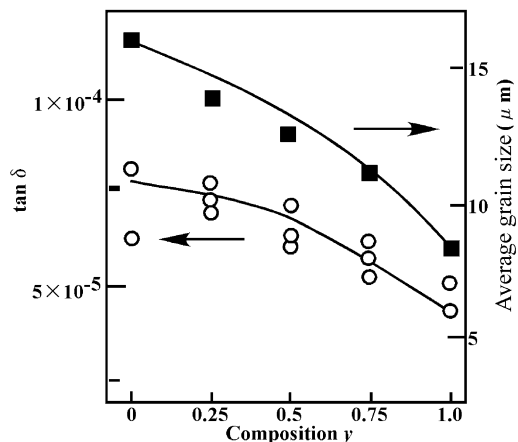


Fig. 5. The  $\tan \delta$  and the average grain sizes of the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$

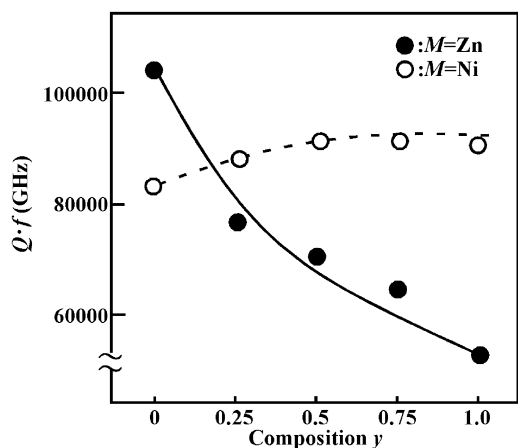


Fig. 6. The  $Q \cdot f$  values of the  $(\text{Mg}_{3.5}\text{M}_{0.5})(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $y=0-1$ ) solid solutions as a function of composition  $y$ .

### 3.3. $(\text{Mg}_{3.5}\text{M}_{0.5})(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$ ( $M = \text{Zn}$ and $\text{Ni}$ ) ( $y = 0-1$ ) solid solutions

In the case of the  $M$  substitution for  $\text{Mg}$ , the  $Q \cdot f$  value of the  $(\text{Mg}_{4-x}\text{M}_x)\text{Nb}_2\text{O}_9$  solid solutions were decreased with increasing the composition  $x$  as mentioned above. However, the  $Q \cdot f$  value of the samples at  $x = 0.5$  are higher than those of the  $\text{RAlO}_3$  and  $\text{R}(\text{Mg}_{1/2}\text{Ti}_{1/2})\text{O}_3$  ceramics.<sup>8,9</sup> Moreover, the  $Q \cdot f$  values of the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions were improved by the partial  $\text{Sb}$  substitution for  $\text{Nb}$ . Thus, the  $(\text{Mg}_{3.5}\text{M}_{0.5})(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $M = \text{Zn}$  and  $\text{Ni}$ ) ( $y = 0$  to  $1$ ) solid solutions were investigated. The  $Q \cdot f$  value of the  $(\text{Mg}_{3.5}\text{M}_{0.5})(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions are shown in Fig. 6. The quality factor of the  $\text{Ni}$ -substituted  $(\text{Mg}_{3.5}\text{Ni}_{0.5})(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions were slightly increased, whereas those of the  $\text{Zn}$ -substituted  $(\text{Mg}_{3.5}\text{Zn}_{0.5})(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions were extremely decreased with increasing the composition  $y$ . The decreases in the  $Q \cdot f$  value of the  $(\text{Mg}_{3.5}\text{Zn}_{0.5})(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  were attributed to the decrease in the density caused by  $\text{Zn}$  vaporization.

## 4. Conclusion

The microwave dielectric properties of the  $(\text{Mg}_{4-x}\text{M}_x)(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  ( $M = \text{Zn}$  and  $\text{Ni}$ ) solid solutions were investigated; the limits of the  $(\text{Mg}_{4-x}\text{M}_x)\text{Nb}_2\text{O}_9$  and the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions were  $x = 2$  and  $y = 1.5$ , respectively. The quality factors of the  $(\text{Mg}_{4-x}\text{M}_x)\text{Nb}_2\text{O}_9$  solid solutions were decreased by the  $M$  substitutions for  $\text{Mg}$ , whereas those of the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions were remarkably increased from 192 268 to 285 423 GHz. It was found that the increase in the  $Q \cdot f$  values of the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions may be related to the grain size decrease which is opposite to general results and this relation is consistent with that of  $\text{Al}_2\text{O}_3$ . Moreover, the partial  $\text{Sb}$  substitution for  $\text{Nb}$  in the  $(\text{Mg}_{3.5}\text{M}_{0.5})(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions was also effective in increasing the  $Q \cdot f$  values. The optimum microwave dielectric properties were obtained for the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions; the properties at  $y = 1$  are:  $\epsilon_r = 10.0$ ,  $Q \cdot f = 285\,423$  GHz and  $\tau_f = -66.2$  ppm/ $^\circ\text{C}$ . The microwave dielectric properties of the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions are comparable to those of  $\text{Al}_2\text{O}_3$ ; it is considered that the  $\text{Mg}_4(\text{Nb}_{2-y}\text{Sb}_y)\text{O}_9$  solid solutions are one of the suitable substrate for the integrate circuits.

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## References

- Alford, N. McN., Breez, J., Wang, X., Penn, S. J., Dalla, S., Webb, S. J., Ljepojevic, N. and Aupi, X., *J. Eur. Ceram. Soc.*, 2001, **21**, 2605–2611.
- Kumada, N., Taki, K. and Kinomura, N., *Mater. Res. Bull.*, 2000, **35**, 1017–1021.
- Kan, A., Ogawa, H. and Yoshida, A., *The 40th Sympo. Basic Sci. of Ceram.*, 2002, 262–263.
- Rietveld, H. M., *J. Appl. Crystallogr.*, 1969, **2**, 65–71.
- Izumi, F., *Rietveld Method*, ed. R. A. Young Oxford University Press, Oxford, 1993, Chapter 13.
- Hakii, B. W. and Coleman, P. D., *IRE Trans. Microwave Theory Tech.*, 1960, **8**, 402–410.
- Penn, S. J., Alford, N. McN., Templeton, A., Wang, X., Xu, M., Reece, M. and Schrapei, K., *J. Am. Ceram. Soc.*, 1997, **80**, 1885–1888.
- Cho, S. Y., Kim, I. T. and Hong, K. S., *J. Mater. Res.*, 1999, **14**, 114–119.
- Cho, S. Y., Kim, C. H., Kim, D. W. and Hong, K. S., *J. Mater. Res.*, 1999, **14**, 2484–2487.